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# Effect of SILAR Cycles on the Thickness, Structural, Optical Properties of Cobalt Selenide Thin Films

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**Abstract:** Cobalt Selenide thin films were fabricated using Successive Ionic Layer Adsorption and Reaction (SILAR) deposition technique at different SILAR cycles. The precursors for Cobalt and Selenium ions were CoCl<sub>2</sub>.6H<sub>2</sub>O and Na<sub>2</sub>SeSO<sub>3</sub> respectively. Optical properties and thickness of the deposited films were studied to determine the effect of number of SILAR cycles on these properties. The optical absorbance of the films was found to decrease as wavelength increases and increases as SILAR cycle increases. Transmittance of the CoSe thin films was found to increase as the wavelength increases but decreases as number of SILAR cycles increased. The extinction coefficient of CoSe thin films decreases as wavelength increases but increases as the SILAR cycles increases. The energy band gap of CoSe thin films deposited decreases from 2.47 eV to 2.20 eV as number of SILAR cycles increases and film thickness increases from 92.96 nm and 225.63 nm. Structural properties of deposited cobalt selenide thin films with crystallite size ranging from 7.63 nm to 13.07 nm.

Keywords: Cobalt selenide, Thin Films, SILAR method, Optical Properties, XRD

### 1. Introduction

Binary transition metal chalcogenide systems have recently attracted unlimited consideration due to the fact that they possessed exceptional physical and chemical properties which open windows for various applications. Their ideal optical and electrical properties contributed to them being widely use in solar energy conversion, optoelectronics industries [1, 2]. Recently, huge interests have been channeled to developing these binary transition metal chalcogenides most especially that of selenide families to determine the possibilities of tailoring their optical properties to suite the required values for maximum solar spectrum. According to [3, 4], selenium based compounds have excellent properties that make them good materials for ionic crystal semiconductors, metal superconductors which include ferromagnetism and ferrimagnetism. The semiconducting selenide of interest in this work is Cobalt Selenide (CoSe) thin films.

Cobalt selenide which is one of the transition metal chalcogenide systems belonging to II - VI semiconductor with direct band gap [5]. It has been

studied by many researchers for various applications such as dye – sensitized solar cell [6], nano devices [7], magnetic dipoles [8], supercapacitor [9], catalyst for oxygen and hydrogen reduction reaction [10, 11, 12] and photocatalytic water oxidation [13]. Cobalt selenide thin films have been grown using deposition methods such as electrodeposition [2, 10, 14], chemical bath method [1, 5, 15, 16,]. There were no available literatures on deposition of CoSe by SILAR method. This singular fact is motivation for this research work.

In this work, cobalt selenide thin films were deposited by SILAR method. Number of SILAR cycles were optimized to determine its effect on the optical and structural properties of the deposited thin film materials.

# 2. Experimental details

The growth of CoSe thin films was carried out by the SILAR method at room temperature. Aqueous solution containing  $[Co(NH_3)_4]^{2+}$  was prepared by mixing 5 mL of 36 % NH4OH with 80 mL of 0.1 M CoCl<sub>2</sub>. 6H<sub>2</sub>O until Co(OH)2 was precipitated. This solution was used as a cationic precursor and was kept at room temperature, while 80 mL solution of freshly prepared Na<sub>2</sub>SeSO<sub>3</sub> was used as anionic precursor. Na<sub>2</sub>SeSO<sub>3</sub> was prepared by refluxing 8 g of selenium powder with 250 ml of 1 M of sodium sulphite at 90 °C for 2 hours. The following procedure was adopted to deposit CoSe thin films.

One SILAR growth cycle with cycle time (t\_c) equals to 80 seconds shown in figure 1 involved four steps which are Immersion of the cleaned substrate in first reaction bath containing solution of 0.1 M Co(OH)<sub>2</sub> for 30s. This process leads to absorption of Co<sup>2+</sup> ions on the surface of the substrate.

The substrate was rinsed with double distilled water (DDW) for 10 s to remove excess  $Co^{2+}$  ions that were loosely bounded to the glass substrate. The substrate was then immersed in freshly prepared Na<sub>2</sub>SeSO<sub>3</sub> solution for the 30 s. The selenide (Se<sup>2-</sup>) ions reacted with the absorbed Co2+ions on the active center of the substrate to form CoSe films. Again, the substrate was rinsed in DDW for 10 s to remove loosely bounded ions present on the substrate and unreacted Co2+ and Se<sup>2-</sup> ions. The procedure was repeated for 5 cycles to get a uniform film with good thickness.

At the end of a cycle, uniform films were seen on the surface of the microscopic glass substrate. Increase in film thickness was observed as cycles increase. The processes mentioned above were repeated for four other samples with increased number of SILAR cycles of 10, 15, 20 and 25 respectively. The distinctiveness of SILAR method lies in the easy control of thickness and other properties of thin film for various device applications. Deposited CoSe thin films were subjected to optical and structural using StellarNet UV – VIS – NIR spectrometer (Blue – Wave Miniature; Model: UVNb) and x – ray diffraction machine (Brucker D8 high resolution X-ray diffractometer) Thicknesses of the thin films were obtained using gravimetric techniques.

# 3. Results and Discussion

#### **3.1. Structural Analysis**

Figure 2 shows diffractograms of cobalt selenide thin deposited at 5 and 25 SILAR cycles. The results showed that x - ray pattern of the deposited thin film corresponding to orthorhombic phase of CoSe with JCPDS file number (53 – 0449). The  $2\theta$  angles shown in table 1 correspond to miller indices of [110], [101], [111], [211], [031]. A new peak at 62.81° was observed when SILAR cycles increased to 25. Increased intensity was observed as SILAR cycles increased from 5 to 25. This showed improvement in the crystallinity as a result of increase in film thickness as SILAR cycles increased. The improvement in the crystallinity could be of the films as SILAR cycles increased. The crystallite size (D) was calculated using Debye - Scherrer's formula in equation (1) as given by [17, 18] while the dislocation density ( $\delta$ ), lattice strain ( $\varepsilon$ ) and inter-planar spacing (d) were evaluated using equation (2), (3) and (4) as given by [19, 20].



**Figure 1** Schematic diagram of SILAR method for the deposition of CoSe thin films (•  $\rightarrow$  Co<sup>2+</sup>, •  $\rightarrow$  Se<sup>2-</sup>):  $t_d \rightarrow$  dip time and  $t_c \rightarrow$  cycle time

$$D = \frac{0.9\lambda}{\beta\cos\theta} \tag{1}$$

$$\delta = \frac{n}{p} \tag{2}$$

$$\varepsilon = \frac{\beta}{\beta}$$
 (3)

$$d = \frac{\lambda}{2\sin\theta} \tag{4}$$

Where  $\beta$  is the full width half maximum (FWHM) of the observed  $2\theta$  angles,  $\theta$  is the diffraction angle, n equals to 1 at minimum dislocation density and  $\lambda$  is the wavelength of Cu – ka radiation (1.54059 A) used in the x – ray diffraction experiment. The crystallite size of the films is between 7.63 nm to 13.07 nm. The slight increase observed showed that the crystalline nature of the films increases with increase in SILAR cycles. Decrease in lattice strain and dislocation density confirmed the improvement in the crystal structure of SILAR deposited cobalt selenide as cycle increases. Similar orthorhombic structural phase of CoSe<sub>2</sub> were obtained by [21, 22, 23].



**Figure 2** Diffractograms of deposited Cobalt Selenide Thin Films at 5 cycles and 25 cycles.

#### 3.2 Thickness Measurement

The thickness of the films shown in figure 3 was obtained using gravimetric method. The area (A) of the substrates covered by the deposited cobalt selenide thin film was measured.by recording dimensions of the substrate covered by the film. The length and width were 4.0 cm and 2.5 cm respectively which gave an area of 11.25 cm<sup>2</sup>. The bulk density ( $\rho$ ) of CoSe material is 7.65 g/cm<sup>3</sup> while the change in mass ( $\Delta m$ ) of the substrates was obtained by measuring the mass of the substrates before and after deposition. Values of thickness (t) of the deposited thin films were evaluated using equation (5) as given by [20, 24, 25].

$$t = \frac{\Delta m}{A\rho} \tag{5}$$

While growth rates  $(R_g)$  were evaluated using equation (6)

$$R_g = \frac{film \ thickness \ (nm)}{number \ of \ SILAR \ cycles} \tag{6}$$

Figure 3 shows the plot of thickness and growth rate of the films against number of SILAR cycles. From the figure, the thickness of the films increases as cycle SILAR increases. Thickness of 92.96 nm, 116.20 nm, 162.67 nm, 209.15 nm and 255.63 nm were obtained for 5, 10, 15, 20 and 25 cycles respectively. Peak thickness 255.63 nm was obtained for film deposited at 25 SILAR cycles while the least thickness 92.96 nm was obtained for film deposited at 5 SILAR cycle. This increase is due to formation of ionic layer as SILAR cycle increases. As expected, in one SILAR cycle a single layer of CoSe<sub>2</sub> film was deposited. As SILAR cycles increase, more layers of CoSe<sub>2</sub> were deposited which leads to increase in film's final thickness This result is similar to increase in the thickness as SILAR cycles increase obtained by [26, 27, 28, 29].

Table 1      Structural Properties of Cobalt Selenide Thin Films						
Samples	<b>2θ (</b> °)	FWHM	d – spacing	D	$\delta  imes 10^{15}$	$\epsilon  imes 10^{-3}$
-	-	(rad)	(Å)	(nm)	(lines/m²)	
5 cycles	23.45	0.0165	3.79	8.94	12.52	19.93
-	30.33	0.0231	2.95	6.50	23.64	21.28
	34.07	0.0164	2.63	9.23	11.73	13.38
	47.18	0.0221	1.92	7.15	19.55	12.65
	53.43	0.0247	1.71	6.56	23.27	12.29
			Average	7.63	18.18	14.83
25	23.45	0.0081	3.79	18.36	2.96	9.7
cycles	30.32	0.0157	2.95	9.54	11.00	14.5
	34.43	0.0076	2.60	19.95	2.51	6.13
	47.18	0.0168	1.93	9.43	11.24	9.59
	53.42	0.0163	1.71	9.95	10.10	8.09
	62.81	0.0152	1.48	11.16	8.03	6.23
			Average	13.07	7.64	9.04

According to [27], the increase maybe as a result of increase in reaction time available for the growth of thin film. Growth rates ( $R_g$ ) of deposited CoSe<sub>2</sub> were found to decreased as SILAR cycles increases. From 15 SILAR cycles, the growth rate decreased gradually from 10.85 nm/cycle to 10.23 nm/cycles. The trend is similar to result obtained by [26].



**Figure 3** Plot of thickness and growth rate against number of SILAR cycles for Cobalt selenide thin films.

#### **3.3 Optical Properties**

The optical study was done within the wavelength ranges of 300 nm to 1100 nm which represent ultraviolet (UV), visible light (VIS) and near infrared (NIR) portion of the electromagnetic radiation respectively. From the absorbance (A) obtained from spectrophotometer, other properties such as transmittance, extinction coefficient and band gap were calculated. Transmittance (T) was determine using equation (7) as given by [30, 31].

$$T = 10^{-A}$$
 (7)

Adsorption coefficient values of the deposited thin film were estimated using equation (8) as given by [32, 33].

$$\alpha = \frac{1}{d} \ln \left( \frac{1}{T} \right) \tag{8}$$

d is the thickness of the deposited thin film and T is the transmittance. Extinction coefficient values of deposited thin film were estimated using equation (9) as given by [34, 35].

$$K = \frac{\alpha\lambda}{4\pi} \tag{9}$$

Where  $\alpha$  is the optical absorption coefficient and  $\lambda$  is the wavelength. Energy band gap values of the films were estimated using equation (10) as given by [36, 37].

$$\alpha h v = \beta \left( h v - E_g \right)^n \tag{10}$$

Where  $\beta$  is an independent constant and n is an index that describes the degree of electron transition by the deposited thin films. In general,  $n = \frac{1}{2}$  for direct allowed transition, and  $n = \frac{3}{2}$  for indirect forbidden transitions. Skin depth ( $\delta$ ) that measured the degree of penetration

of photons within the surface of the deposited cobalt selenide thin films was estimated using equation (11) as presented by [38, 39].

$$\delta = \frac{1}{\alpha} \tag{11}$$



**Figure 4** Plot of Absorbance against Wavelength for Cobalt selenide thin films.



**Figure 5** Graph of Transmittance plotted against wavelength for Cobalt selenide thin films.

Figure 4 shows the graph of absorbance of the films plotted against wavelength. From the graph, absorption peak was observed for all the films at 385 nm. This showed that there is no shift in the absorption peaks of CoSe thin films as number of SILAR cycle increases. The absorbance values were found to decreases as wavelength increase. Also, the graph revealed an increase in absorbance as number of SILAR cycles increases. This increase in the absorbance values of CoSe thin film may be as a result of increase in the amount of CoSe atoms formed as the number of cycle increases. Increase in absorption of radiation is due to increase in film thickness as SILAR cycles increased from 5 cycles to 25 cycles. The absorbance values of the films ranged between 10.89 % at 1100 nm to 111.33 % at 300 nm. This result shows that CoSe thin films absorbed UV light more than VIS and NIR.



**Figure 6** Graph of Extinction coefficient plotted against wavelength for Cobalt selenide thin films.

Figure 5 revealed the graph of transmittance of the films plotted against wavelength. From the graph. The transmittance of CoSe thin films increases as wavelength increases. Also, the graph showed a decrease in transmittance as number of SILAR cycles increases. The transmittance values of the films decrease from 300 nm to minimal values at 385 nm before increasing steadily to peak values at 1100 nm. Transmittance values of the films at 300 nm are 33.35 %, 26.03 %, 19.17 %, 16.15 % and 12.88 %. Transmittance values of the deposited thin films at 385 nm are 25.32 %, 18.57 %, 13.16 %, 10.22 % and 7.70 % while transmittance values at 1100 nm are 77.81 %, 73.53 %, 69.05 %, 65.94 % and 62.61 % respectively.

Figure 6 shows the graph of extinction coefficient of the films plotted against wavelength. From the graph, extinction coefficient peak was observed for all the films at 385 nm. The extinction coefficient of CoSe thin films decreases as wavelength increase. Also, the graph shows an increase in extinction coefficient as number of SILAR cycles increases. Extinction coefficient values of the films at 300 nm are  $3.79 \times 10^{-2}$ ,  $4.65 \times$  $10^{-2}$ ,  $5.60 \times 10^{-2}$ ,  $6.30 \times 10^{-2}$  and  $7.08 \times 10^{-2}$ . Peak values at 385 nm are  $4.75 \times 10^{-2}$ ,  $5.82 \times 10^{-2}$ ,  $7.01 \times$  $10^{-2}$ ,  $7.88 \times 10^{-2}$ , and  $8.86 \times 10^{-2}$  while the values at 1100 nm are  $8.67 \times 10^{-3}$ ,  $1.06 \times 10^{-2}$ ,  $1.28 \times 10^{-2}$ ,  $1.44 \times 10^{-2}$  and  $1.62 \times 10^{-2}$ . This result confirmed the absorptive nature of the deposited CoSe<sub>2</sub> thin films when exposed to ultraviolet A (UVA) radiations and the films are less absorptive when visible light and near infrared radiations passed through them.



Figure 7 Graph of  $(ahv)^2$  plotted against photon energy (eV) for Cobalt selenide thin films.

Figure 7 shows the plotted graphs of  $(\alpha hv)^2$ against photon energy (hv) for different number of SILAR cycles carried out. The direct band gap energy was extrapolated at the axis of the photon energy (hv)where  $(\alpha hv)^2 = 0$ . CoSe thin film deposited at 5 SILAR cycles has energy band gap of 2.40 eV. CoSe thin film deposited at 10 SILAR cycles has energy band gap of 2.35 eV. CoSe thin film deposited at 15 SILAR cycles has energy band gap of 2.30 eV. CoSe thin film deposited at 20 has band gap of 2.25 eV. CoSe thin film deposited at 25 SILAR cycles has band gap of 2.10 eV. This show that the energy band gap of CoSe thin films deposited decreases as number of SILAR cycles increases. The values of energy band gap obtained is between the values of 1.5 eV and 3.75 eV obtained [40, 41]. The energy band gap values obtained in this work are in line with values of 2.40 eV to 3.60 eV obtained by [15]. Our energy band gap result shows a reduction in band compared to values of 4.30 eV to 4.25 eV obtained by [5].





**Figure 8** Graph of Energy band gap plotted against thin film thickness for Cobalt selenide thin films.

**Figure 9** Graph of Skin depth against Photon energy (eV) for Cobalt selenide.

Figure 8 shows the variation of energy band gap with number of SILAR cycles. The energy band gap is found to decrease as film thickness increases. This shows the possibility of tuning the band gap of CoSe thin films at different band parameters. According to [29], the decrease in energy band gap due to increase in SILAR cycles that resulted to increase in film thickness may be as a result of quantum confinement effect, enhancement of the crystallization of the films at lower SILAR cycles and variation in the stoichiometry of the formed thin films. According to [42], energy band gap decreased with increasing SILAR cycles and this is due to increase in the thickness of the films which resulted to increase in the density of the localized state in the energy band gap of CoSe thin film.

Figure 9 showed the plot of skin depth against photon energy for deposited cobalt selenide thin films. Skin depth as an important thin film property defines the absorption of photons within the thin film surface. The skin depth was found to decrease as SILAR cycles increases. Cut – off photon energy ( $E_{cut-out}$ ) of 3.315 eV and corresponding cut – off wavelength ( $\lambda_{cut-off}$ ) of 375.28 nm was obtained for the SILAR deposited cobalt selenide thin films. Beyond this cut – off wavelength, the absorption effect disappeared and the amplitude goes into extinction as light propagates through a large distance.

#### 4. Conclusion

Successive ionic layer adsorption reaction (SILAR) method has been used to successfully deposited cobalt selenide thin films using cobalt (II) chloride hexahydrate, distilled water and sodium selenosulphate obtained by refluxing selenium powder with sodium sulphite. The deposited thin films were subjected to optical and structural characterization using spectrophotometer and x - ray diffractometer. Absorbance results show that the deposited cobalt selenide thin films have absorption peak at 385 nm. There was no shift in the absorption peak to either longer or shorter wavelength rather the absorbance of the films increases as number of SILAR cycles increase and decreased as wavelength increases with maximum absorbance of 1.11 (111 %) obtained at 385 nm for films deposited with 25 SILAR cycles and least absorbance of 0.11 (10.89 %  $\approx$  11 %) obtained within the NIR region. Transmittance results of the SILAR deposited cobalt selenide revealed an increase in the transmittance as wavelength increases from 300 nm to 1100 nm. Also, the transmittance decreases as number of SILAR cycles increases. The extinction coefficient of the deposited films decreases as wavelength increases and increases as number of SILAR cycles increases. Peak extinction coefficient values of  $4.75 \times 10^{-2}$ ,  $5.82 \times 10^{-2}$ ,  $7.01 \times$  $10^{-2}$ ,  $7.88 \times 10^{-2}$  and  $8.86 \times 10^{-2}$  were obtained at 385 nm for 5, 10, 15, 20 and 25 SILAR cycles respectively, while least values of  $8.67 \times 10^{-3}$ ,  $1.06 \times 10^{-2}$ ,  $1.28 \times$  $10^{-2}$ ,  $1.44 \times 10^{-2}$  and  $1.62 \times 10^{-2}$  were obtained at 1100 nm. Energy band gap of the deposited cobalt selenide thin film falls between 2.20 eV and 2.47 eV. The energy band gap was found to decrease as number of SILAR cycle increases. This decrease in band gap is as a result of increase in the thickness of the films which is due to increase in the number of SILAR cycles carried out. This shows that cobalt selenide thin films are slightly wide - band gap material. These energy band gap values obtained correspond to great extent with result obtained by other researchers. The thickness of the films is between 92.96 nm and 225.63 nm.

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#### **Conflict of interest**

The authors have no conflicts of interest to declare that they are relevant to the content of this article.

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