



Yttrium Doped MoO₃ Nanoplates as Photocatalysts for Enhanced Removal of Organic dyes in Aqueous Solutions

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Abstract: The release of organic dyes into the environment is a significant concern, as they are toxic and carcinogenic, and contribute to water pollution, posing health risks to all living beings. Effective and environmentally friendly approaches such as photocatalysts are essential to deal with organic dyes in industrial wastes. The affordable and environmentally friendly properties of MoO₃ make it a superior photocatalyst compared to other materials. In this study, explores a co-precipitation method was used to synthesize undoped and Yttrium-Doped MoO₃ nanoparticles and its analyzed using various characterization techniques. From structural analysis revealed that all the studied samples were exhibited a monoclinic structure. SEM images showed uniform nanoplate like morphology with reduced particle agglomeration at optimal doping levels which indicates that Yttrium doping concentration affects the morphology. The performance of Yttrium doping on MoO₃ nanoparticles as efficient photocatalyst was further demonstrated in the degradation of various pollutants such as MB, CV and Rh B under visible light irradiation and it shows well photocatalytic activity during the degradation of Rh B under visible light. It was found that an appropriate amount of Y³⁺ dopant can greatly increase photocatalytic activity and the sample with 7 wt. % of Y³⁺ doping exhibits the highest photocatalytic efficiency at 91.5% for Rhodamine B after 140 min of exposure to visible light. Based on kinetic analyses, the photocatalytic performance of the 7 wt.% Y³⁺ doping MoO₃ sample showed four times higher efficient than that of the undoped MoO₃. The results highlight how controlled Y³⁺ incorporation can dramatically improve the photocatalytic properties of MoO₃, positioning 7 wt.% Y³⁺: MoO₃ as a highly promising material for environmental remediation applications.

Keywords: Yttrium Doped MoO₃, Photocatalysis, Methylene Blue, Rhodamine B, Crystal Violet, UV-Visible Light

1. Introduction

Advanced oxidation processes (AOPs), particularly those based on photocatalytic green technologies, have emerged as promising alternatives to conventional wastewater treatment methods due to their remarkable effectiveness in degrading toxic dyes [1, 2]. Among these, semiconductor photocatalysts have attracted considerable attention for their efficiency in eliminating hazardous dye pollutants [3]. Extensive examine has been devoted to understanding and optimizing photocatalytic reactions, where light energy triggers chemical transformations that break down complex dye molecules into simpler, less harmful compounds.

Photodegradation-a key mechanism in photocatalytic processes-involves the decomposition of dye molecules under the influence of sunlight or

ultraviolet (UV) light. Upon exposure to light, the molecules absorb energy, undergo excitation, and subsequently alter their chemical structure through oxidation or reduction reactions. The efficiency of this process hinges on several factors intrinsic to the photocatalyst, such as a high surface area, a wide energy band gap, and the efficient generation of electron-hole pairs. A larger surface area allows more dye molecules to adsorb onto the catalyst, thereby enhancing photodegradation activity.

Numerous semiconductors have been investigated for their photocatalytic potential, including WO₃, ZnO, In₂O₃, and SnO₂, particularly when modified with dopants to improve performance [4-7]. Among them, molybdenum trioxide (MoO₃) has emerged as a promising n-type semiconductor due to its wide band gap (2.8–3.6 eV), high photoactivity, and responsiveness to both UV and visible light [8]. Its

exceptional physical and chemical properties have led to its widespread application across diverse fields such as lithium-ion batteries [9], catalysis [10], field emission [11], gas sensing [12], and optoelectronic devices including photochromic and electrochromic systems [13,14], as well as in electrode materials [15], capacitors [16], and ferromagnetic materials [17].

One strategy to further enhance the properties of MoO₃ involves the introduction of dopants, which create surface defects that can adapt and control the material's functionality. These defects not only influence the electronic band structure but can also induce spectral shifts in the UV region, thus broadening the material's photo reactivity [18]. Rare-earth (RE) dopants are known to demonstrate interesting catalytic properties, high surface basicity, and fast oxygen ion mobility which are believed to be important in photocatalytic applications [19]. Notably, the low optical phonon energy of MoO₃ compared to other metal oxides makes it particularly suitable for rare-earth (RE) doping, which can exploit its variable oxidation states [20].

Among RE elements, yttrium ions (Y³⁺) have shown exceptional promise as dopants. Due to their 4d orbital occupancy, Y³⁺ ions introduce multiple electron configurations, which in turn modify the energy band structure, enhance the surface-to-volume ratio, and create active grain boundary centers [21]. These modifications make Y³⁺ a highly effective dopant for improving the photocatalytic behaviour of host materials like MoO₃.

In the present study, Pure and Y³⁺-doped MoO₃ nanoparticles were synthesized with varying dopant concentrations (3, 5 and 7 wt.%) and assessed for their structural, optical, and photocatalytic properties. Specifically, the doped MoO₃ demonstrated degradation efficiencies of approximately 92% for Rhodamine B (Rh B), 64% for Methylene Blue (MB), and 30% for Crystal Violet (CV) dyes. These results confirm the potential of Y³⁺-doped MoO₃ as a highly efficient photocatalyst for wastewater treatment applications under UV-visible irradiation.

2. Experimental Procedure

In pursuit of synthesizing high-performance photocatalysts, the study employed reagent-grade chemicals, which were further purified whenever necessary to ensure optimal reaction conditions. The preparation of both pure molybdenum trioxide (MoO₃) and yttrium-doped MoO₃ (Y³⁺-MoO₃) was carried out using a co-precipitation method, with dopant concentrations systematically varied at 0, 3, 5, and 7 wt.%.

The synthesis of pure MoO₃ began with a stoichiometric amount (0.01 mol) of ammonium heptamolybdate tetrahydrate [(NH₄)₆Mo₇O₂₄·4H₂O] (AHM), dissolved in 100 mL of deionized water to form a

clear solution. To initiate the reaction, concentrated hydrochloric acid (37% HCl) was gradually introduced to the solution. This mixture was then transferred to a sand bath and heated at 150 °C for two hours. During this period, the reaction progressed, leading to the formation of a colloidal suspension with a solid precipitation. After the reaction was completed, the temperature was slowly reduced to room temperature, allowing the precipitate to settle. The resultant product was then thoroughly washed and centrifuged in three cycles using distilled water and ethanol to remove residual impurities [22]. The washed precipitate was dried overnight in an oven at 80 °C. To achieve complete crystallization and remove any remaining moisture or organic residues, the dried powder underwent calcination at 450 °C in a muffle furnace for one hour. This procedure yielded phase-pure MoO₃ powder, suitable for structural and photocatalytic analysis.

For the preparation of Y³⁺ doped MoO₃ samples, a similar co-precipitation route was adopted. In this case, Solution A was prepared using the same concentration of AHM (0.01 mol) dissolved in 100 mL deionized water. Parallely, Solution B was freshly prepared by dissolving 0.005 mol of yttrium nitrate hexahydrate Y(NO₃)₃·5H₂O in another 100 mL of deionized water. The Y³⁺ doping levels 3, 5, and 7 wt.% were precisely controlled relative to the amount of MoO₃ in order to explore the influence of dopant concentration on material performance.

After mixing the solutions, the resultant powder was manually ground for 30 minutes to ensure homogeneity. The final step involved sintering the mixed powders at 450 °C using a muffle furnace through three sequential thermal cycles, promoting dopant integration and crystal development. This meticulous synthesis process enabled the formation of Y³⁺ incorporated MoO₃ nanoparticles with enhanced structural and photocatalytic properties, setting the stage for subsequent characterization and application testing.

3. Characterization

Implementation of X-ray diffractometer (Shimadzu-LabX XRD-6000, Japan) with CuK α radiation (40 kV, 200 mA, $\lambda = 1.540568$) at 10°-80° of 2 θ angle allows the identification of prepared samples' phases. NEL-SEM 4800 operates at 200 kV while analyzing the nanoparticle surface morphology and elemental components under observation at 200 kV. JEOL JEM 2100 high-resolution transmission electron microscope allows for bright and dark field imaging, lattice resolution, selective region electron diffraction at 200 kV acceleration voltage, and image recording with a Gatan orious CCD camera. Raman examinations used mini raman Microprobe Imaging system 3000 which combined with an optical microscope offering 1.5 μ m resolution power and 2 cm⁻¹ spectral resolution capability. Shimadzu 3600 UV-visible irradiation spectrophotometer was used to record the changes in

different dyes solutions. A photocatalytic evaluation took place for pure and MoO₃ materials containing different yttrium doping levels (0, 3, 5 and 7 wt.%).

4. Results and Discussion

4.1 X-Ray Diffraction Analysis

The XRD analysis of co-precipitation synthesized pure MoO₃ and Y³⁺ doped MoO₃ nanoparticles with concentrations from 0 to 7 wt.% appears in Figure. 1 (a). The monoclinic phase of MoO₃ nanostructure occurred through well intense diffraction peaks positioned at 2θ degrees of 23.7, 26.0, 27.6, 34.2, 39.4 and 49.0 which correspond to the crystalline planes (111), (-311), (310), (220), (-113) and (004) in pure and Y³⁺ doped MoO₃ samples. Analysis showed that the obtained peaks of monoclinic phase MoO₃ matched the reference data stored in JCPDS card number 078-1078. The analysis of pure and Y³⁺ doped MoO₃ patterns indicated that no secondary reflections existed for the 0, 3, 5 and 7 wt. % doping samples. A new reflection (*) shows up as a peak at 31.6° in the XRD pattern of the 7 wt.% Y³⁺: MoO₃ nanoparticles is denoted in the Fig. 1(a). The substitution of Y³⁺ occurs at the sub-lattice's locations throughout the molybdenum trioxide matrix structure. Y₂Mo₄O₁₅ exists as a supplementary phase in this material [23]. The low intensity of diffraction from Y₂Mo₄O₁₅-phase reveals that it exists as an unwanted impurity substance among the main MoO₃-phase. XRD peak intensities increase when the yttrium concentration in the doping process reaches from 0 to 7 wt.%. The distinct pronounced peak proves that the synthesized materials exhibit high crystallinity quality. A high degree of crystallinity exists in the 7 wt.% doped sample when compared to its three counterparts. The XRD patterns

displayed in Figure. 1 (b) show an expansion of the data between 26 and 30 degrees. The (310) peak positions of doped MoO₃ microstructures move to lower angles in comparison to pure MoO₃ structures. The increased peak shift demonstrates that dopants have been successfully integrated into the MoO₃ crystal framework without forming separate phases. Physical attributes of Y³⁺ (Y³⁺= 1.04 Å) ions lead to PXRD peak displacement because their radius (1.04 Å) surpasses Mo⁶⁺ (0.56 Å). Higher radii of dopants compared to Mo will result in peak shifts to lower 2θ readings whereas smaller radii result in the opposite effect [24, 25]. The extent of shifting indicates how well dopant elements successfully integrate into the MoO₃ crystal network without forming separate phases. The analysis confirms Y³⁺ atoms efficiently replace Mo ions in MoO₃ host structure due to larger atomic radius (Y³⁺) dopant in the MoO₃ lattice instead of Mo ions corresponding to structural distortions [26]. The crystallite size (D), calculated using Debye-Scherrer's formula [27].

$$D = \frac{k\lambda}{\beta \cos \theta} \quad (1)$$

Where, K is the Scherrer constant (0.94), λ is the wavelength of X-ray used (1.54Å), β is the full width at half maximum (FWHM) and θ is the Bragg's diffraction angle. The calculated structural parameters are showed in Table 1. Analysis indicated the crystallite size of the samples ranged from 18.31 nm to 52.90 nm and 43.29 nm and 41.17 nm in this order. The studies demonstrated that Y³⁺ dopant concentration acted as a factor which determined the crystallite dimensions. XRD spectra demonstrated that pure and Y³⁺ doped MoO₃ formed correctly and revealed the dopant density caused changes to both structural attributes and crystallite dimensions of the resulting host material.

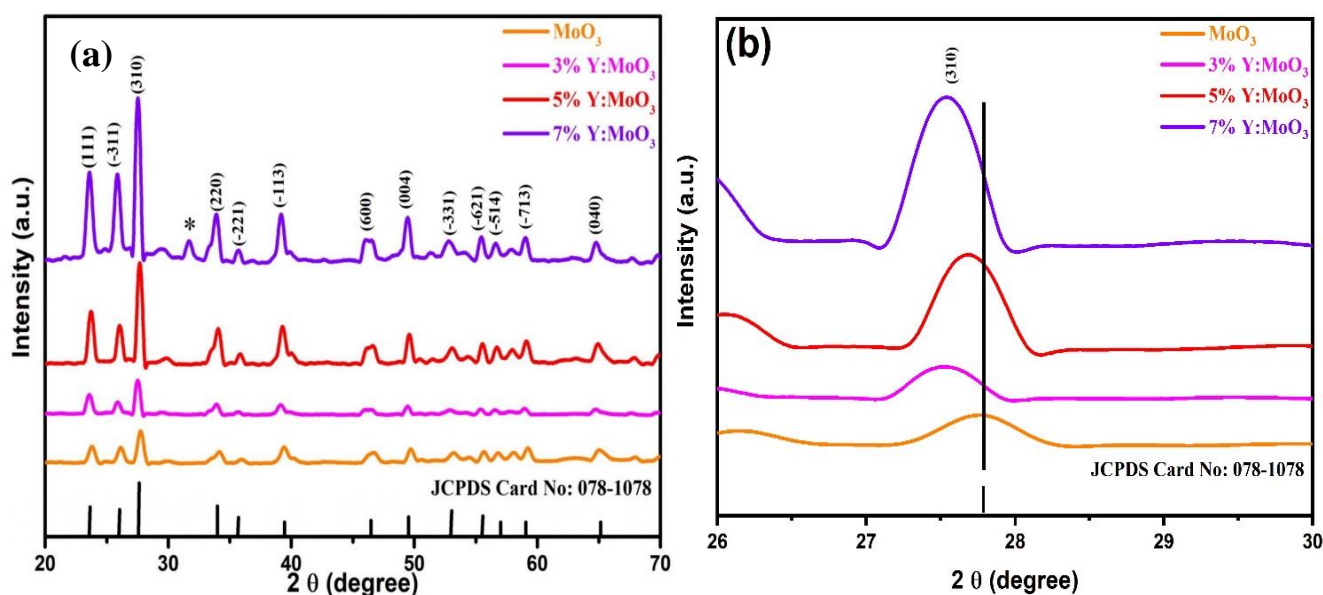


Figure1. (a) XRD analysis of pure and Y³⁺: MoO₃ (3, 5 and 7 wt.%) nanoparticles, (b) Magnified image of peak (310)

Table 1. To calculated for the structural parameters of Pure and Y³⁺ doped MoO₃ nanoparticles for the XRD peak (310)

Samples	FWHM (β)	Grain Size (D)
MoO ₃	0.4349	18.3
Y ³⁺ : MoO ₃ (3%)	0.1506	52.9
Y ³⁺ : MoO ₃ (5%)	0.1840	43.2
Y ³⁺ : MoO ₃ (7%)	0.1840	41.1

4.2 FE-SEM

To gain insights into the surface morphology and structural evolution of the synthesized materials, scanning electron microscopy (SEM) was employed to analyze both pure and Y³⁺ doped MoO₃ nanoparticles at a nanometric scale. High-resolution SEM imaging at 100,000× magnification allowed detailed observation of morphological features, especially at the 100 nm scale. The SEM image of pure MoO₃, shown in Figure. 2(a), revealed a network of uniformly distributed, multiplex-shaped nanoplates measuring approximately 45 μm in length and 175 nm in width. These nanoplates were well-defined, forming a relatively consistent architecture. However, upon doping with Y³⁺ ions, subtle yet significant transformations in morphology were observed across the varying concentrations of 3, 5, and 7 wt.%.

At 3 wt.% Y³⁺ doping, as seen in Figure. 2(b), the material maintained its nanoplate-like structure, though a mild degree of agglomeration became evident. These agglomerates were randomly oriented and varied in size, suggesting that the dopant subtly disrupted the

uniformity of the original structure. Notably, the Y³⁺ doped MoO₃ exhibited more consistency in shape and size compared to pure MoO₃, indicating enhanced growth control introduced by the dopant. As the concentration increased to 5 wt.%, the morphological changes became more pronounced. Figure. 2(c) highlights the emergence of a more regular substructure within the nanoplates, which displayed improved uniformity and definition. This structural refinement likely stems from the influence of Y³⁺ ions on the nucleation process, promoting a more ordered growth pattern. Interestingly, this level of doping also marked the point at which the nanocrystal size began to decrease, indicating a transition in growth dynamics.

When Y³⁺ doping reached 7 wt.%, the resulting MoO₃ structures, illustrated in Figure. 2(d), continued to exhibit the nanoplate formation but with evident changes in their dimensions. Specifically, nanorod lengths began to shrink, leading to a reduction in aspect ratio. Although the nanoplate structure was preserved, increased doping led to slight growth in overall particle size, likely due to the dopant's interference with the crystallization pathway. The observed decrease in particle dimensions throughout the samples can be attributed to the restricted nucleation and altered surface diffusion dynamics that dominate the growth mechanism. A key factor influencing these changes lies in the ionic radii mismatch between Mo⁶⁺ (0.65 Å) and Y³⁺ (0.90 Å). This size disparity prevents Y³⁺ from effectively substituting into the MoO₃ lattice, leading to local lattice distortions that impede crystallite growth. As a result, the introduction of Y³⁺ introduces structural disorder, altering the thermodynamic and kinetic factors governing nanoplate formation.

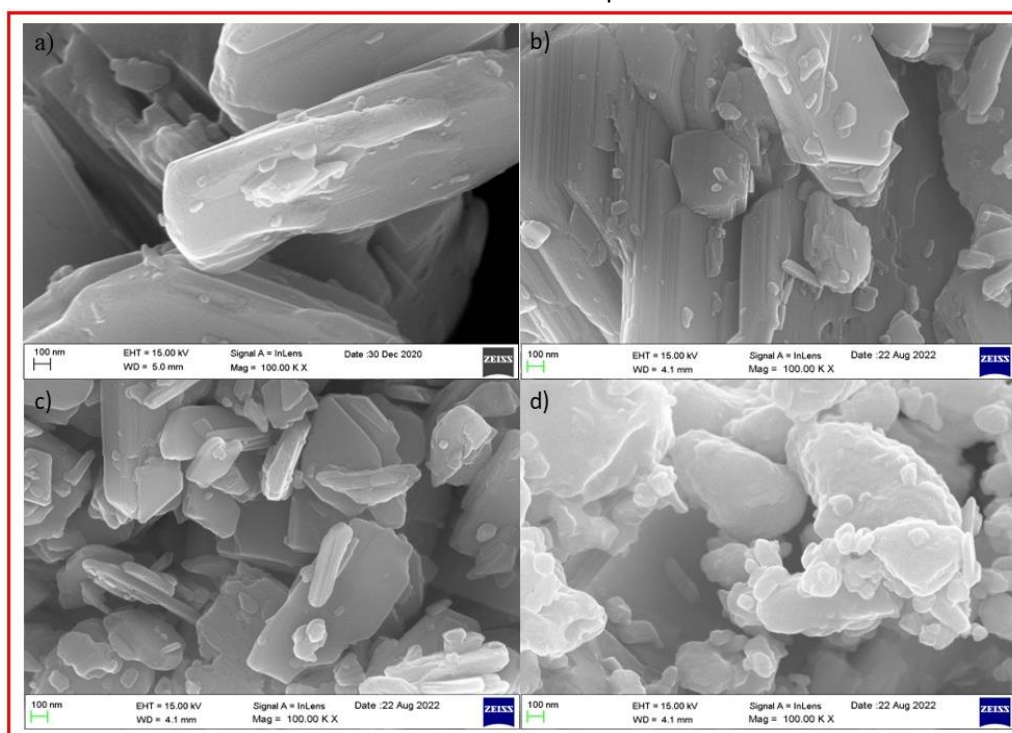


Figure 2. FESEM images of (a) Pure MoO₃ (b-d) Y³⁺: MoO₃ (3, 5 and 7 wt.%) nanoparticles

Further SEM analysis confirmed the presence of morphology in pure MoO₃ nanoparticles (~30 nm in diameter) is sized nanoplates. This observation aligns with earlier findings by [28], who reported a similar mixed morphology in pure MoO₃ synthesized via precipitation. With Y³⁺ doping, however, the dominant morphology shifted toward finer nanoparticles with dimensions shrinking below 20 nm (Figure. 2(b–d)). The dopant appeared to function as a growth inhibitor, suppressing the natural development of large crystalline grains and promoting size reduction through lattice distortion [29].

Increased Y³⁺ content also led to a rise in bimodal particle density, with coarse-grained structures becoming more prominent. These coarser features coexisted with smaller impurities that became embedded in the voids between nanoplates, enhancing the material's packing density. The interaction between differently sized particles generated a wedging effect driven by electrostatic forces, contributing to the formation of a bimodal dispersion as supported by earlier studies [30]. Among the different concentrations examined, the 7 wt.% Y³⁺: MoO₃ sample stood out with superior nanoplate uniformity and optimized width and length dimensions. This suggested an ideal balance between dopant-induced refinement and preservation of structural integrity, establishing it as a potentially optimal composition for further photocatalytic performance assessments.

4.3 EDAX Analysis

The Yttrium doped sample identification takes place through EDAX analysis. EDAX spectra present the compositional analysis of pure and doped samples annealed at 450°C in Figure. 3 (a–d). The acquired spectrum indicates that the sample consists of Mo, O, and Y³⁺ at trace amounts. The Table 2 provides information about elemental weight percentage and atomic percentage content according to their compositional distribution. The intensity reduction of the Mo peak indicates Y³⁺: MoO₃ formation along with an atomic percentage decrease of oxygen. The photocatalytic properties of Y³⁺ doped MoO₃ samples develop because of the present deficiencies as shown by observation.

4.4 TEM analysis

Transmission electron microscopy (TEM) was used to analyze the structure at 200 nm and 10 nm scales, providing information about particle dimensions and morphology, as shown in Figure. 4(a–d). FESEM results appear in the TEM images which show pure MoO₃ contains nanoparticles together with few nanoplates. Most of the MoO₃ nanoplates display uniform three-dimensional structures. The pure MoO₃ nanoparticles exhibited irregular particles with sharp edges according to previous investigate [31].

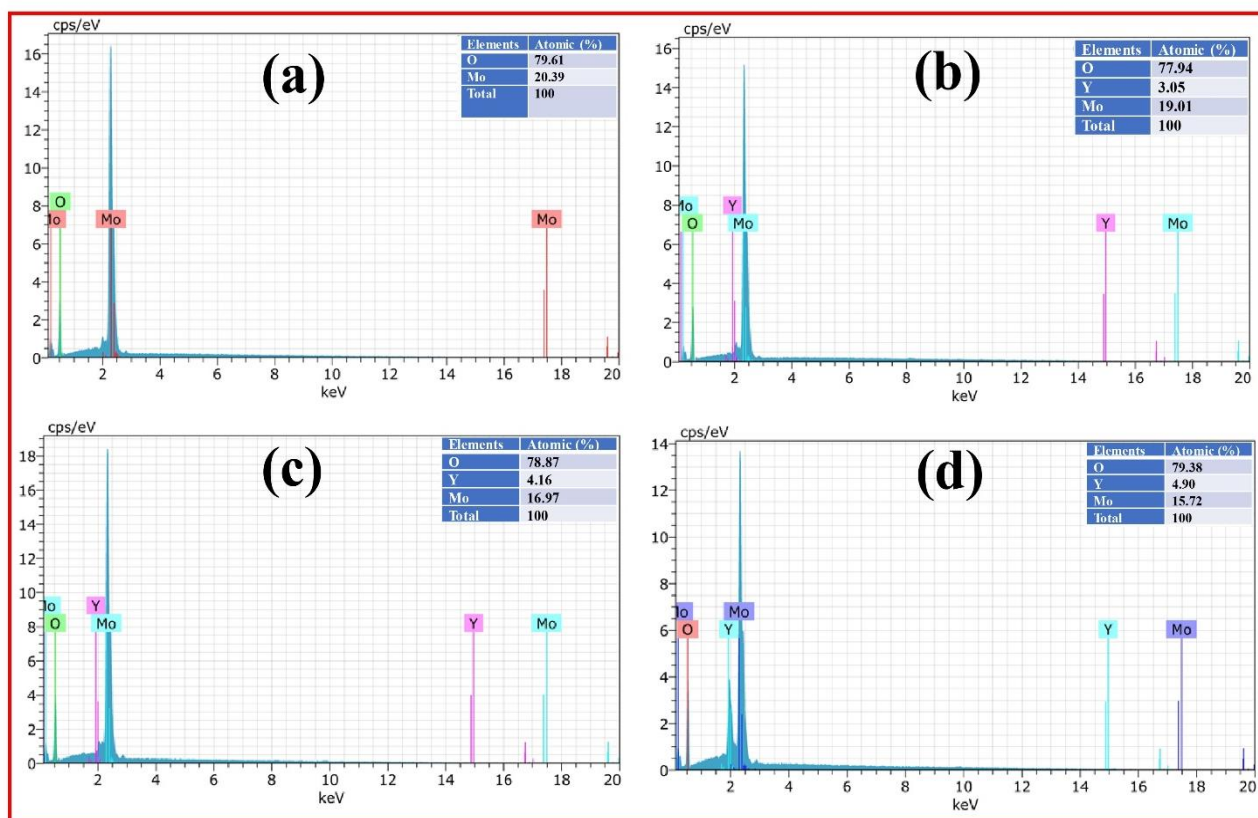
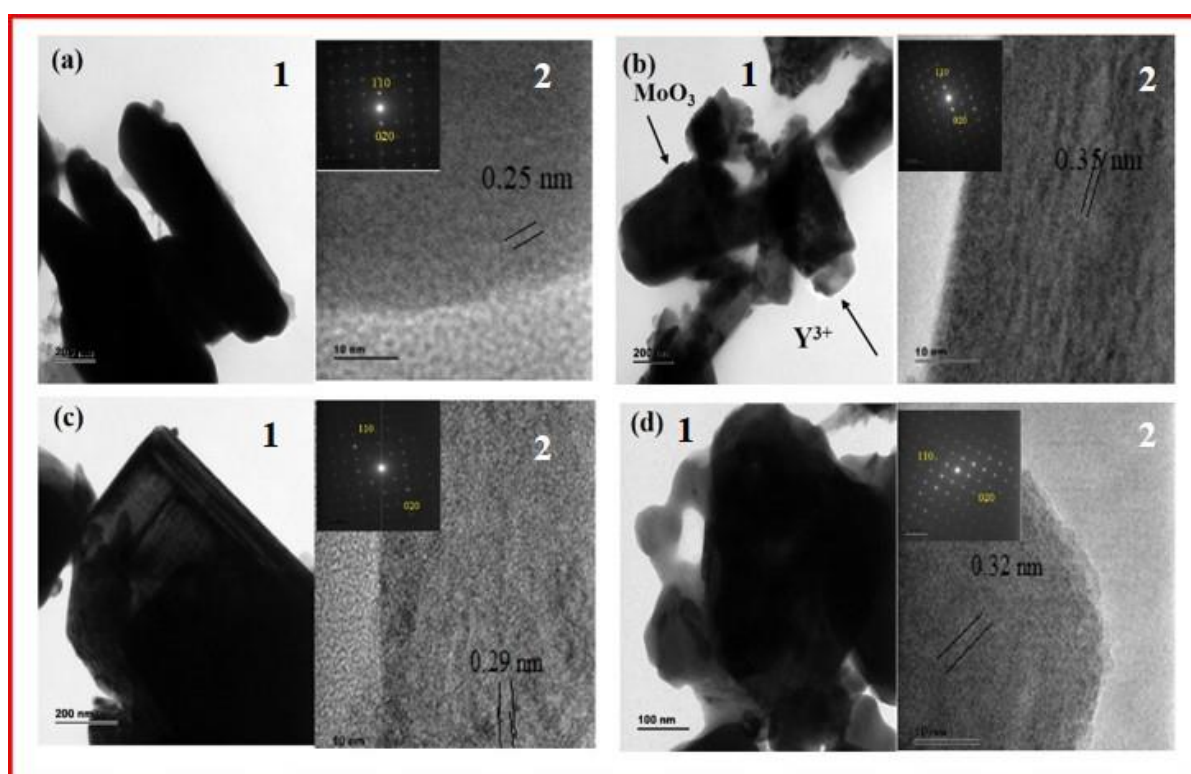


Figure 3. EDAX images of (a) pure and (b-d) Y³⁺: MoO₃ (3, 5 and 7 wt.%) nanoparticles

Table 2. comparison the three dyes for the Maximum Degradation efficiency of Pure and Y³⁺: MoO₃ (3, 5, 7 wt.%) nanoparticles

Samples	Degradation Efficiency (%) MB Dye	Degradation Efficiency (%) Rh B Dye	Degradation Efficiency (%) CV Dye
MoO ₃	63.33	78.33	54.08
Y ³⁺ : MoO ₃ (3%)	46.04	91.5	66.6
Y ³⁺ : MoO ₃ (5%)	62.1	86.7	64.1
Y ³⁺ : MoO ₃ (7%)	64.0	88.4	29.5

**Figure 4.** TEM images of (a) Pure MoO₃ with insert SAED Pattern and Lattice Fringes, (b) Y³⁺ doped MoO₃ wt. at 3% with SAED Pattern and Lattice Fringes, (c) Y³⁺ doped MoO₃ wt. at 5% with SAED Pattern and Lattice Fringes, (d) Y³⁺ doped MoO₃ wt. at 7% with SAED Pattern and Lattice Fringes

The shape evolution of synthesized MoO₃ irregular particles led to agglomerated irregular particles when yttrium ions were present during growth. The structure of MoO₃ particles became agglomerated irregular after the addition of 3 wt. % Y³⁺ doped MoO₃. Finally, varied Y³⁺ amounts led to minimal particle shape modifications. The micrographs of 3 wt. % Y³⁺ doped MoO₃ showed a severe morphological change due to the excessive growth of Y³⁺: MoO₃ on irregular-particle MoO₃. Higher contents of Y³⁺ doped MoO₃ show clusters in TEM images while the particles exhibit reduction in their dimensions. FESEM and XRD results confirm the nanoparticle growth that Y³⁺ doping induces in the material. The recording and presentation of SAED

patterns through HR-TEM images and inserted figures occurred for every analysed material. The distance between lattice planes evolves differently depending on the amount of Y³⁺ dopant according to HRTEM images. The pure MoO₃ sample showed a lattice spacing value of 0.37 nm (020) among its structure. XRD data revealed two crystals spacing of 0.37 nm (020) from pure MoO₃ together with a second spacing of 0.35 nm (111) in the 3, 5 and 7 wt.% Y³⁺ doped MoO₃ samples which signify the appearance of Y³⁺: MoO₃ phase [32]. The TEM images confirm formation of pure and yttrium doped MoO₃ material for all doping levels including 0, 3, 5 and 7 wt.%. Each Y³⁺ increase step led to a successive material formation process.

4.5 Raman studies

Figure 5(a-d) displays the typical Raman spectra of pure MoO_3 and Y^{3+} : MoO_3 sample with three concentrations 0, 3, 5 and 7 wt. %. There are four main peaks in the 283, 665, 817 and 995 cm^{-1} frequency ranges that appear in the Raman spectra. All tested samples show Raman peaks that identify MoO_3 structure with intensity located at 817 cm^{-1} . Explore published by multiple investigations [33, 34] validates the position of these primary Raman peaks that appear in monoclinic MoO_3 crystalline phase. The bending (deformation), twisting and stretching vibrational modes of the material appear within distinct frequency ranges: between 100 and 600 cm^{-1} and between 600 and 1000 cm^{-1} respectively. The peak assignments follow the results of the single-crystal study published by Py *et al* [35]. The O=Mo=O wagging vibration (O=Mo=O wagging) displays its twist mode at 283 cm^{-1} while the stretching mode of triply corresponding oxygen (O–Mo–O stretch) exists at 665 cm^{-1} . The stretching mode of terminal oxygen (Mo=O stretch) manifests at 995 cm^{-1} . Within this peak range Mo=O stretch of Mo2-O (Mo=O stretch) reaches 817 cm^{-1} . As Y^{3+} concentration grows the stretching mode of vibration together with lattice mode diminishes. They indicates that band intensity grows stronger while the temperature elevation occurs. Raman scattering depends on how crystalline materials are within the spectrum because crystalline materials produce sharp intense peaks, but amorphous materials lead to less intense broader peaks. Modern studies demonstrate that the Raman spectra between crystalline and amorphous or semicrystalline solids (having identical chemical makeup) display significant variations due to differences in spatial order and long-range translational symmetry presence or absence [36]. Every peak identified in the Raman spectra belongs specifically to Pure MoO_3 nanoplates and Y^{3+} doped MoO_3

nanoparticles examined separately. The different relative peak intensities of both the lattice mode and Mo=O terminal oxygen stretching mode show evidence that structural distortion and oxygen defects occur post Y^{3+} introduction into MoO_3 nanoplates.

4.6 Photocatalytic Studies

Deposition testing of pure and Yttrium doped materials exposed three dyes to Methylene Blue (MB), Rhodamine B (Rh B) and Crystal Violet (CV). The degradation process of MB Rh B and CV dyes occurred through the observation of peaks at 665 nm, 554 nm and 590 nm. Testing for degradation efficiency of the samples occurred intermittently during twenty-minute periods. The absorption signal intensities drop in a controlled manner when the irradiation time extends.

Investigators studied the MB dye breakdown course for pure MoO_3 and Y^{3+} : MoO_3 (3, 5 and 7 wt. %) crystals when using visible light irradiation. Visible light exposure of MB dye resulted in a continuous decrease of its chemical compound amounts. The visible light irradiation duration led to distinct changes in MB dye absorbance which the Figure 6 (a) depicts. The degradation rate for MB dye through visible light radiation reached 63.33%, 46.04%, 62.1% and 64.0% based on Y^{3+} : MoO_3 nanoparticle concentrations of 0, 3, 5 and 7 wt.%. The sample containing 7 wt. % MoO_3 nanoparticles exhibits the greatest photocatalytic effect among all other Y^{3+} : MoO_3 systems tested. The MoO_3 degradation performance improves through Y^{3+} doping concentrations and porous nanoplate formation that provides increased dye absorption capacity because of their abundant surface area [44]. The presence of high oxygen-deficient properties in Y^{3+} : MoO_3 nanoparticles enabled their enhanced catalytic action.

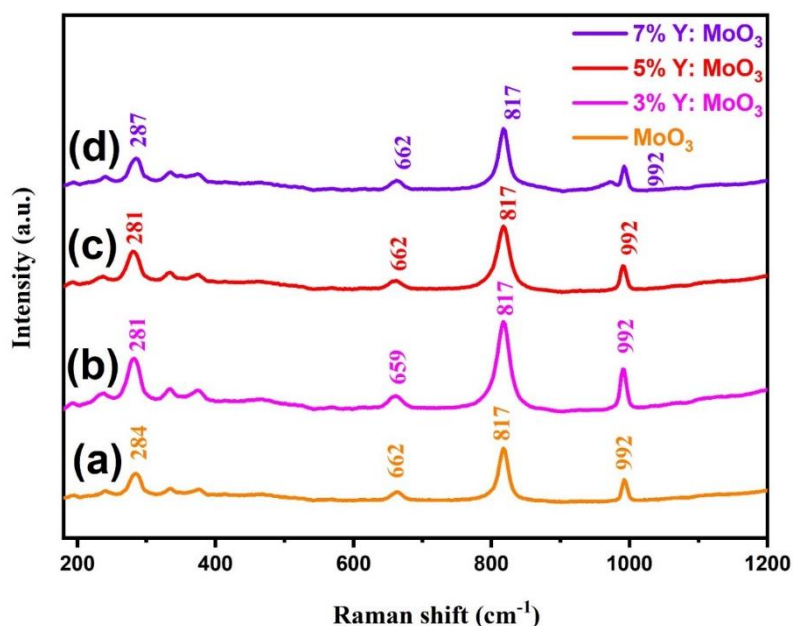


Figure 5. Raman spectrum of (a) pure and (b-d) Y^{3+} : MoO_3 (3, 5 and 7 wt.%) nanoparticles

Methylene Blue

(a)

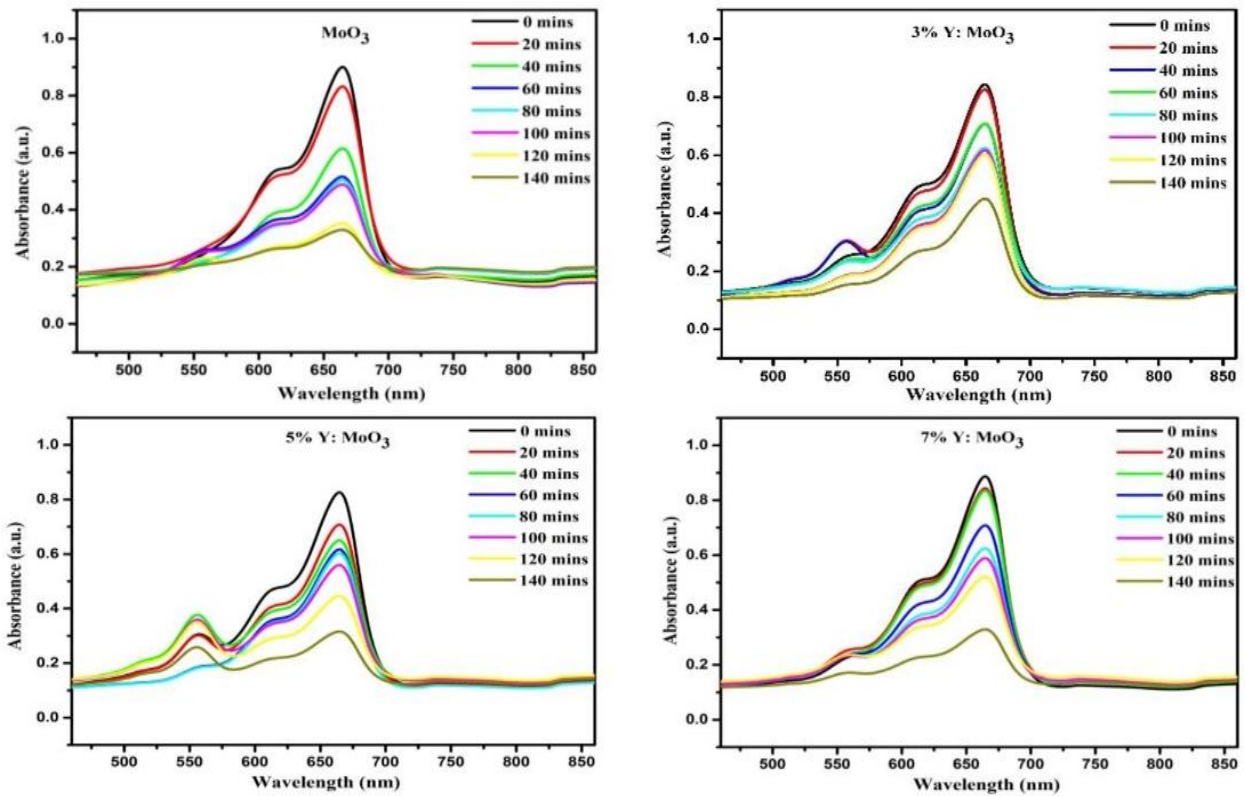


Figure 6. (a) Photocatalytic activity of Pure and Y^{3+} : MoO_3 (3, 5 and 7 wt.%) nanoparticles in Methylene Blue dye

Rhodamine B

(b)

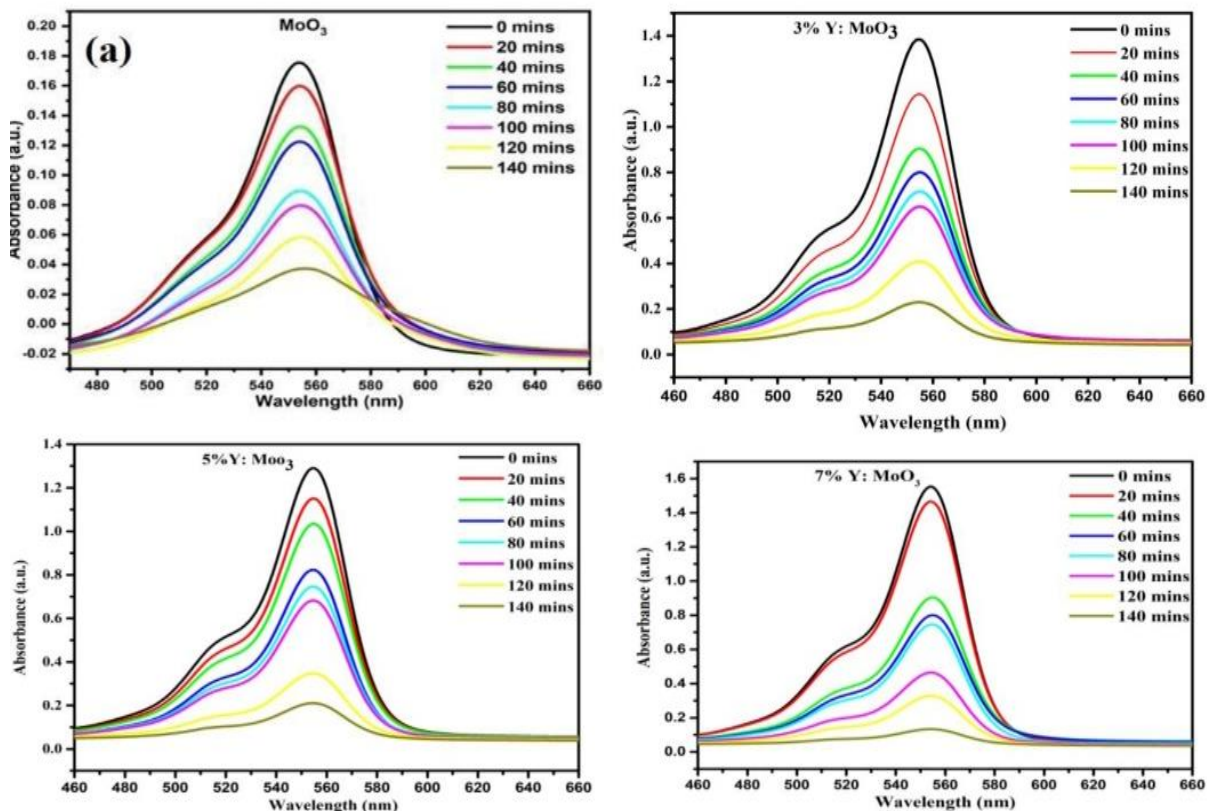


Figure 6. (b) Photocatalytic activity of Pure and Y^{3+} : MoO_3 (3, 5 and 7 wt.%) nanoparticles in Rhodamine B dye

(c) **Crystal Violet**

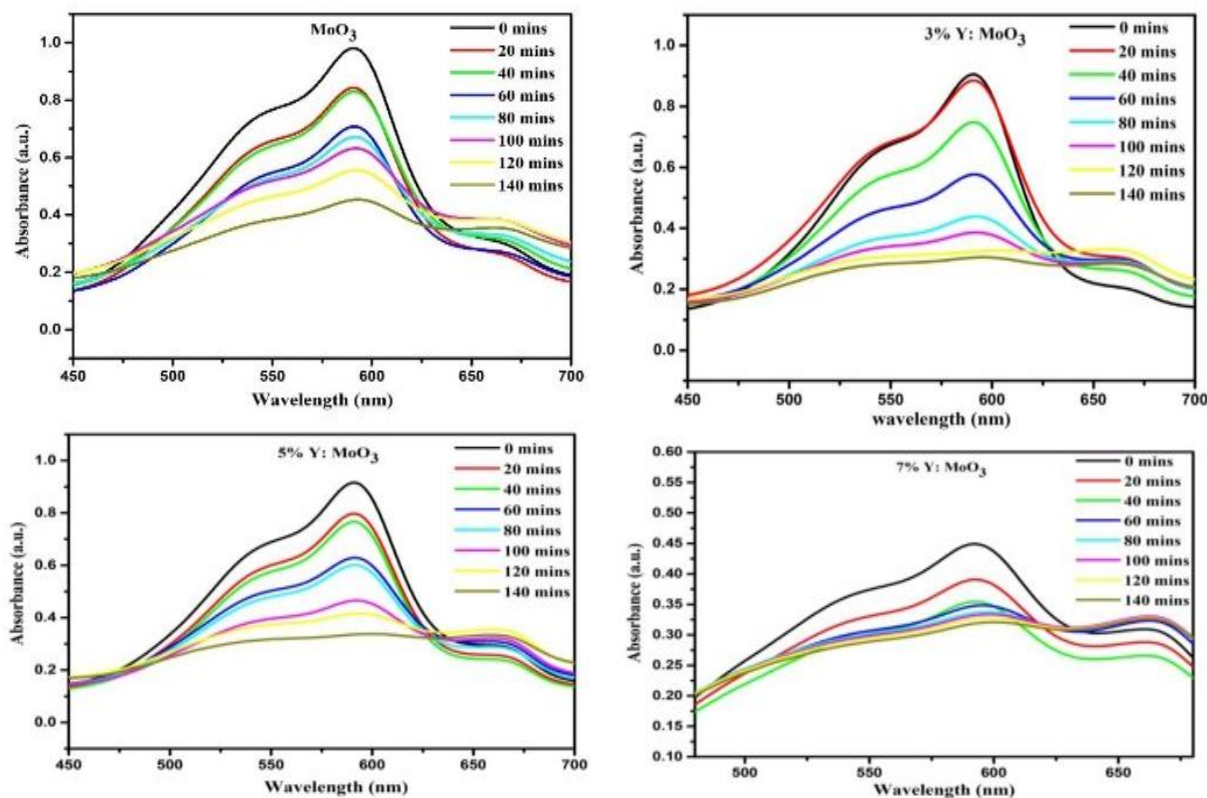


Figure 6. (c) Photocatalytic activity of Pure and Y³⁺: MoO₃ (3, 5 and 7 wt.%) nanoparticles in Crystal Violet dye

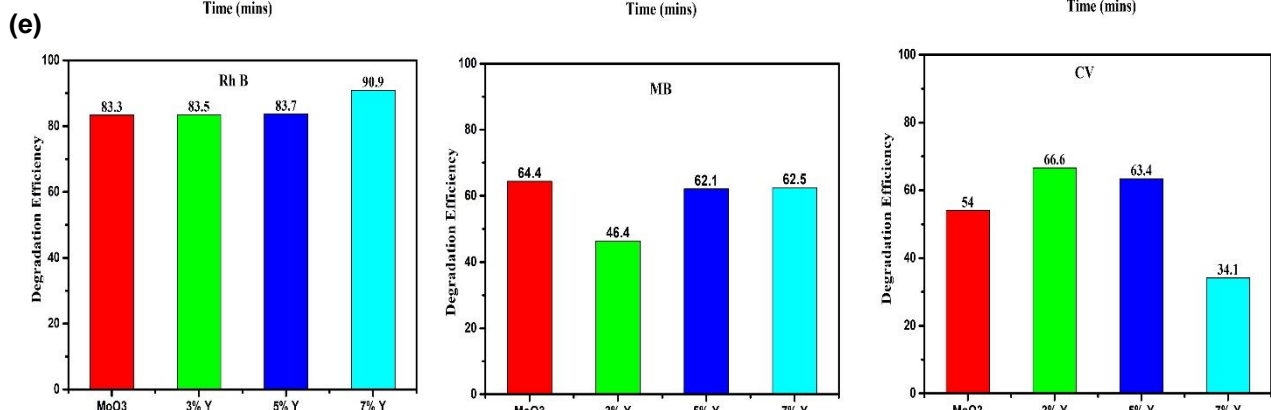
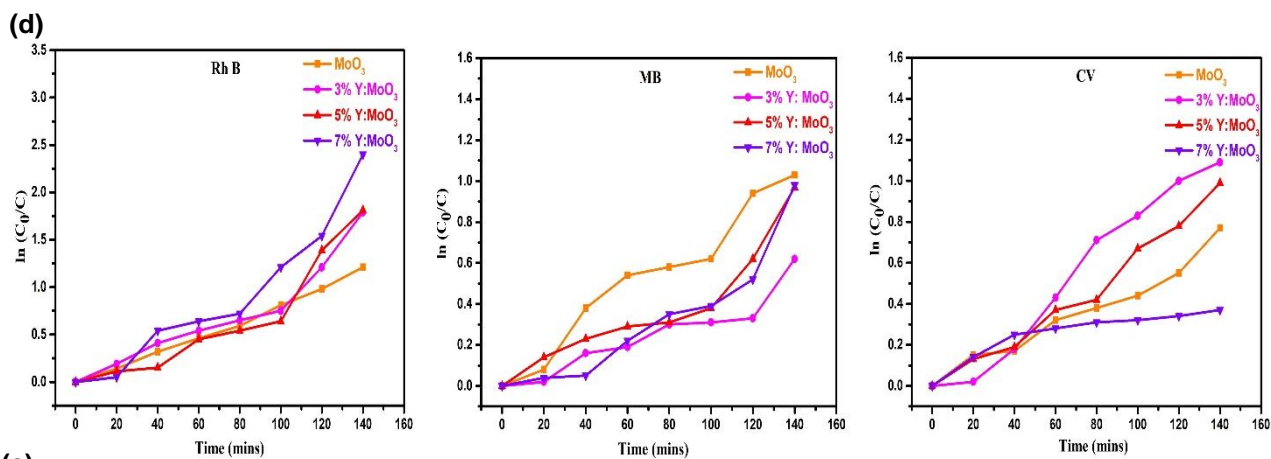


Figure 6. (d) The photocatalytic kinetics curve of pure and Y³⁺: MoO₃ nanoparticles (e) degradation efficiency of MB, Rh B and CV dyes degradation under visible light irradiation.

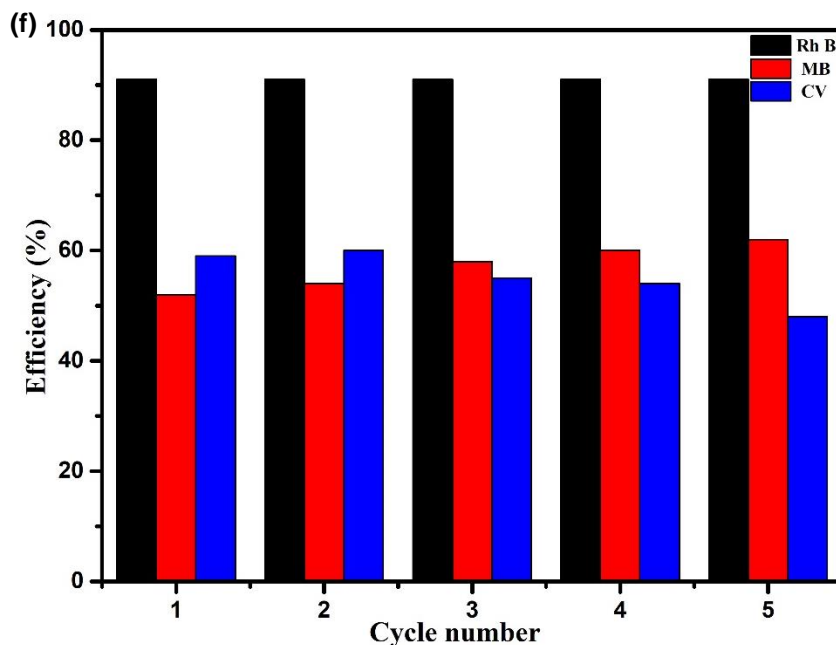


Figure 6 (f). Cycling photodegradation test of Y^{3+} : MoO_3 (7 wt. %) nanoparticles with different organic dyes

Under visible light exposure for 20 minutes the 7 wt.% Y^{3+} : MoO_3 nanoparticles achieved 50% dye color reduction that surpassed the other prepared materials which reached only 15% according to Figure 6 (b). The 7 wt.% Y^{3+} : MoO_3 sample succeeded in degrading Rh B dye by 91.5% under one hour of radiation exposure to become the most effective sample with degradation exceeding 50% for all other studied compositions. The Rh B photodegradation process reached its peak with the 7 wt.% Y^{3+} : MoO_3 sample which improved electron hole recombination while its band gap surpassed the pure and other concentration levels. The XRD and FE-SEM evaluation results indicate that materials with greater surface area enable increased molecule attachment which enhances their performance for photoactivity-related water splitting operations and catalytic reactions [38, 39].

The concentration of CV dye lowers as it interacts with visible light. The UV-vis absorption spectra show how CV dye reacts through visible light radiation at different measurement points. The degradation efficiency of CV dye reached 54% for pure MoO_3 nanoparticles but increased to 66% for Y^{3+} : MoO_3 (3 wt.%) and corresponded to 64% for Y^{3+} : MoO_3 (5 wt.%) while Y^{3+} : MoO_3 (7 wt.%) showed 29% degradation efficiency. The photocatalytic performance of Y^{3+} : MoO_3 (3 wt.%) nanoparticles surpasses all other examined samples [38]. Photoactivity became stronger in Y^{3+} doped MoO_3 because Y^{3+} served in the MoO_3 lattice structure to enable successful carrier charge separation. The increased oxygen deficiency in the Y^{3+} : MoO_3 (7% wt.%) nanoparticles contributes to better photocatalytic performance of the material.

The degradation rate constant (k) measurement of three dyes uses the pseudo-first-order kinetic model and shows up in the equation. During the degradation

process the pollutant starts at a low initial concentration. The reaction rates for photocatalytic degradation appear in Fig. 6 (d). The light irradiation time against $\ln(C_0/C)$ provides data for prepared photocatalysts in this depiction. Plots depicting color degradation percentages during degradation time can be found in the same figures. To determine the dye degradation rate following the Langmuir- Hinshelwood relationship is used Eqn. (2) [41-43].

$$\ln(C_0/C) = kt \quad (2)$$

k represents the rate constant whereas the reaction time equals t which measures the initial dye concentration (C_0) against concentration C of three dyes. The degradation level achieved its best outcome with MoO_3 material reaching 91.5 % dye destruction rate together with rate constant 0.0406 min^{-1} while MoO_3 sample displayed minimal degradation at 78.33% along with rate constant 0.0144 min^{-1} throughout 140 minutes of light exposure. The Y^{3+} : MoO_3 (7 wt.%) nanoparticles demonstrate the best rate constant based on visible light irradiation testing making them an outstanding photocatalyst (Table 3).

The decolorization efficiency of the pure sample reaches 78% after took my measurements. At the same time, efficiency differences have decreased to values smaller than 1%. The reduction rate of Y^{3+} doped MoO_3 occurs more radically along with a faster pace. When yttrium rare-earth is added to MoO_3 the system achieves better efficiency when illuminated by ultraviolet light. The decolorization efficiency outcome relies on the quantity of utilized material. They would be incomplete without analyzing how three dyes react to visible light decolorization. The depiction of MB, Rh B and CV degradation efficiency measurements throughout irradiation duration can be observed in Figure. 6 (e).

Table 3. Stakeout the pseudo-first order rate constants (K) Pure and Y³⁺: MoO₃ (3, 5, 7 wt.%) nanoparticles with three cationic dyes

Materials	Rate constant (K) Min ⁻¹			Correlation coefficient (R ²)		
	Rh B	MB	CV	Rh B	MB	CV
MoO ₃	0.0144	0.0165	0.0128	0.8753	1.0030	0.7781
3% Y ³⁺ : MoO ₃	0.0300	0.0102	0.0180	1.0823	0.6200	1.0942
5% Y ³⁺ : MoO ₃	0.0306	0.0160	0.0168	1.8601	0.9726	1.0249
7% Y ³⁺ : MoO ₃	0.0406	0.0168	0.0057	2.4680	1.0212	0.3460

The dye degradation process became faster when the Y³⁺ element was doped into MoO₃. The dye degradation using 7 wt.% Y³⁺: MoO₃ as a photocatalyst occurred at four times the rate of MoO₃ making it suitable for possible photocatalytic usage. The removal performance achieved during the last stage of testing was equal for 7 wt. % Y³⁺: MoO₃ but this sample showed slightly better results than others. Yttrium element addition to molybdenum trioxide particles dramatically enhances the operation speed for removal procedures. The visible light illumination promotes higher activity in all samples compared to what they achieve under UV conditions. Therefore, the longer the wavelength the better the removal efficiency. Several causational factors possibly determined the end results of degradation obtained in this process. The e⁻/h⁺ recombination acts as the most significant operational challenge that reduces photocatalytic degradation performance. The principal objective of doping is to slow down charge recombination while rare-earth ions function effectively at trapping electrons so doping with these ions proves suitable for improving MoO₃'s photocatalytic behavior. It demonstrates that Y³⁺ ions in MoO₃ effectively reduce recombination across the matrix structure. They identified Y³⁺ doped composites as having the most optimal general photocatalytic activity among all doping variants. The second important factor originates from the behavior of the grain microstructure. Particle size stands as a vital parameter which determines the activity in photocatalysis processes [40]. The small-sized particles during photo sensitization expand the specific surface area which facilitates degraded molecule adsorption on the catalyst surface. This is in good agreement with SEM micrographs and XRD analysis. Also, I should note that that the enhancement of the photocatalytic activity could be related to a doping-induced alteration of the MoO₃ band-gap toward higher wavelengths. Since their simultaneous effect, it is very difficult to differentiate the separate contributions of the above-mentioned factors to the overall degradation procedure, and therefore, a full analysis of results is compulsory. Figure. 6 (f) shows the photocatalytic activity efficiency for 7 wt.% Y³⁺: MoO₃ samples over 5 cycles in the degradation of three dyes over a period of 140 minutes.

Especially, ~ 91.5 % degradation of Rh B is still achieved after five cycles of photodegradation, only 7 wt.% Y³⁺: MoO₃ sample is observed in higher efficiency when compared to other two dyes. On the other hand, 7 wt.% Y³⁺: MoO₃ the efficiency of MB is 64 % and third dye the efficiency of 7 wt.% Y³⁺: MoO₃ sample is around 30 %. Lower efficiency of the CV dye with 7 wt.% Y³⁺: MoO₃ sample could be attributed to MoO₃ photocorrosion phenomenon during photocatalytic reactions [44, 45]. On the other hand, doping alkali metal elements can increase the photocorrosion resistance by increasing the chemical stability of MoO₃ during photocatalytic activity [46]. The above results demonstrated that the morphology of MoO₃ nanomaterials played an important role in the photocatalytic property, and more efforts still need to be done on how to enhance the photocatalytic activity of MoO₃ nanomaterials and clarify the photocatalytic mechanism to facilitate the application on pollutants degradation.

This paper explained the exceptional photocatalytic behavior of pure and yttrium doped MoO₃ nanoparticles with various concentrations by comparing their data with previously published studies about three dyes (Rh B, MB, CV) as well as Pure MoO₃ and yttrium doped MoO₃ nanoparticles sample degradations (Table 4 [47-60]).

Finally the photocatalytic degradation efficiencies of three dyes Methylene Blue (MB), Rhodamine B (Rh B) and Crystal Violet (CV) in pure Y³⁺: MoO₃ (3, 5 and 7 wt.%) samples. Dye degradation reached > 91.5% for three organic substances when exposing them to visible light for 140 minutes of irradiation time. Tests indicate that Y³⁺: MoO₃ (7 wt.%) nanoparticles hold potential for visible light degradation of Rhodamine B dye.

Table 4. Comparative performance of photocatalytic activity Pure and Y³⁺: MoO₃ (3, 5, 7 wt.%) nanoparticles with different dyes

S. No	Materials	Concentration (%)	Dye	Light source	Maximum Mins	Dye degradation (%)	Ref.
1	MoO ₃	10	MB	UV light	180	13	[47]
	Y: MoO ₃					88	
2	MoO ₃	15	MB	UV-Visible light	90	96	[48]
	Ni-MoO ₃					77	
3	MoO ₃	5	MB	Visible light	5 hrs	18	[49]
	Zn-MoO ₃					43.6	
4	MoO ₃	15	MB	Visible light	100	96.2	[50]
	Zn-MoO ₃					96.8	
5	MoO ₃ -CdS		Rh B	Visible light	180	97	[51]
6	AgBr/MoO ₃	1:3	Rh B	Visible light	5	95	[52]
7	MoO _{3-x}	88 mmol	MO	Visible light	120	95.4	[53]
				NIR light		76.5	
8	C ₃ N ₄ /MoO ₃	20% wt	Rh B	Visible light	180	93	[54]
	g-C ₃ N ₄ /MoO ₃				120	71.8	
9	MoO ₃		MB	UV-Vis Light		91	[55]
	MoO ₃ /CoFe ₂ O ₃		Rh B			54	
	MoO ₃ /CoFe ₂ O ₃		CV			65	
	MoO ₃ /CoFe ₂ O ₃						
10	CeO ₂ -Y ₂ O ₃	pH - 9.0	Rh B	UV light	210	52	[56]
11	MoO _{3-x} Ag/MoO _{3-x}	65%wt	Rh B	Visible light	12	54	[57]
12	α-MoO ₃		MB	Visible light	50	99	[58]
13	MoO ₃		Rh B	Visible light	80	90	[59]
14	α-MoO ₃		Rh B MB	Visible light	90	95 95	[60]
15	MoO ₃	3%	Rh B MB CV	UV-Visible Light	140	78.3 63.3 54.0	Present work
	Y ³⁺ : MoO ₃		Rh B MB CV			88.4 46.0 66.6	
	Y ³⁺ : MoO ₃		Rh B MB CV			86.7 62.1 64.1	
	Y ³⁺ : MoO ₃		Rh B MB CV			91.5 64.0 29.5	

5. Conclusion

A co-precipitation process served to manufacture MoO₃ nanoplates for UV-visible light photocatalysis through the utilization of Yttrium concentration variations. The analysis through XRD patterns shows that all prepared MoO₃ along with Y³⁺: MoO₃ samples adopt a monoclinic crystal structure. Controlled observation through FE-SEM reveals that all prepared samples have nanoplate like structure. The 7 wt. % Y³⁺: MoO₃ sample stood out with superior nanoplate uniformity and optimized width and length dimensions. The MoO₃ crystal displayed a nearly nanoplate structure according to HR-TEM measurements and it achieved its maximum average size among all emulsions within the sample containing 7 wt. % Y³⁺: MoO₃. The Raman spectroscopic investigation verifies the stable monoclinic phase of the MoO₃ following doping since it matches XRD data. The XRD and FT-IR examination reveals how strong functional groups enter the other samples when using 7 wt. % yttrium doped MoO₃ nanoparticles. The high efficiency of Rh B dye degradation achieved 91.5% proves that 7 wt. % Y³⁺: MoO₃ is superior to other dyes and other samples tested. 7 wt. % Y³⁺: MoO₃ promotes the photocatalytic action of other MoO₃ powder by providing higher active sites because of its small crystals with great surface area. The photocatalytic decomposition of Rh B molecules mostly involves hole participation as the main reactive species.

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Authors Contribution Statement

T. Usha Devi: Writing – original draft, Validation, Software, Resources, Methodology, Investigation, Formal analysis, Data curation, D. Senthil Kumar: Formal analysis, Writing, Review and Editing. M. Gowtham-Formal analysis, Writing, Review and Editing. Senthil Kumar Nagarajan: Conceptualization. Validation, Resources, Methodology, Formal analysis, supervision and Writing – review & editing. All the authors read and approved the final version of the manuscript.

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Competing Interests

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Data Availability

The data supporting the findings of this study can be obtained from the corresponding author upon reasonable request.

Has this article screened for similarity?

Yes

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